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DYNAMIC MIXED MODE FRACTURE

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ABSTRACT

A newly developed data reduction process was used to reevaluate dynamic photoglastic results and to extract dynamic stress intensity factors, $K_{1}^{\rm dyn}$ and $K_{11}^{\rm dyn}$, associated with curved and branched cracks in fracturing Homalite -100 plates. A branching stress intensity factor approximately 5 times the fracture toughness was identified for this material. Moderate to severe crack curvings were associated with a $K_{1}^{\rm cyn}$ and $K_{11}^{\rm cyn}$ ratio as low as 0.05, but with positive remote stress component, $\sigma_{\rm ox}$.

INTRODUCTION

Two dimensional dynamic photoelasticity has been used by Dally and his associates [1,2] and by the author and his associates [3,4] to determine experimentally the dynamic stress intensity factor $K_{\rm L}^{\rm T}$, surrounding a propagating crack and to establish a dynamic fracture toughness, $K_{\rm L}^{\rm T}$, versus crack velocity å, relation which may control dynamic fracture. This use of dynamic photoelasticity in studying dynamic fracture has been eloquently described by J. W. Dally in his recent article [1]. As noted by Dally, the data reduction procedure used by most investigators in the past for calculating $K_{\rm L}^{\rm T}$, from the transient isochromatics surrounding the propagating crack tip, was restricted to Mode I crack tip deformation. A theoretical, near-field, static isochromatics was first equated to the recorded experimental dynamic isochromatics and the resultant static stress intensity factor of the former was considered the dynamic stress intensity factor of the latter [2,5]. Error estimates for using a near-field stress to extract the Mode I dynamic stress intensity factor have been made by several investigators [6-8] and in particular, exhaustively by Rossmanith and Irwin [8].

Studies of the static isochromatic patterns under mixed mode loading conditions, i.e. in the presence of combined $K_{\rm I}$ and $K_{\rm II}$ crack tip deformations were made by C. W. Smith [9,10], Gdousto and Theocaris [11] and more recently by Dally and Sanford [12,13] and Rossmanith [14]. These mixed mode isochromatics are all characterized by their unsymmetric patterns with respect to the straight crack line. It is also interesting to note that the shapes of these static isochromatics are strongly influenced by the higher order terms, i.e. terms other than $K_{\rm I}$ and $K_{\rm II}$. In particular, the second order term of $\sigma_{\rm CX}$, commonly referred to as the remote stress component, will distort the symmetry of the isochromatics by significant stretching and

shortening of the upper and lower loop system [14]. For a pure mode II crack tip deformation, the isochromatic loop straddles the crack tip as shows in Figure I where a nearly pure shear state of stress is generated around branched cracks. The mode II stress intensity factors $K_{I\,\bar{I}}$, and remote stress components $\sigma_{\rm ox}$, associated with these isochromatics are listed in the following Table 1.

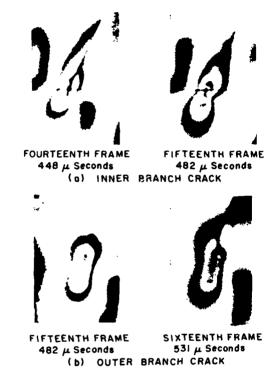


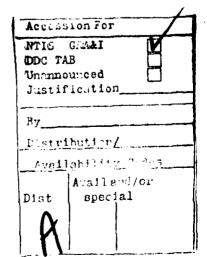
Fig. 1 Typical Mode II Dynamic Isochromatic Parterns of Arresting Branched Cracks. Homalite-100 Single Edge Notched Specimen Under Fixed-Grip Loading. Specimen No. B5

Table 1. K_{II} and σ_{ox} for Arrested Branch Cracks in Fig. 1

(a) Inner Branch Crack

K 11 Oox			Frame MPa√m MPa	frame MPa√m MPa
(b)	Outer	Branch	Crack	
			_	

	15th frame	16th	frame
κ^{11}	0.44 MPa√m	0.41	MPa√n
OX II	0.18 MPa	0.08	MPa



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With the development of a data reduction procedure for evaluating $K_{I\,I}$ together with $K_{I}^{Q\,Y}$ values, it became possible to investigate experimentally the role of mixed mode dynamic near field stresses in dynamic fracture. The authors used such procedure to evaluate the stress intensity factors associated with crack branching and crack curving [16]. The purpose of this paper is to use this data reduction procedure to further extract $K_{I}^{Q\,Y}$ and $K_{I\,I}^{Q\,Y}$ from the previously recorded dynamic isochromatics surrounding running crack tips of curved and branched cracks.

DATA REDUCTION PROCEDURE

A three parameter, mixed mode, near-field state of stresses surrounding a crack propagating at constant velocity [17,18] was used to derive a relation between the Modes I and II dynamic stress intensity factors, $K_{\underline{I}}^{U}$ and $K_{\underline{I}}^{U}$, and the remote stress component $\sigma_{\underline{I}}$, and the dynamic isochromatics. This relation together with an overdeterministic, least-square method formed the basis of a data reduction procedure for extracting the three dynamic parameters $K_{\underline{I}}^{U}$ and $K_{\underline{I}}^{U}$ and $\sigma_{\underline{I}}^{U}$ from the recorded dynamic photoelastic pattern surrounding a running crack. Further details of this data reduction procedure can be found in Reference [16].

Figure 2 shows two frames out of a 16-frame dynamic photoelastic record of a curving crack in a notch bend specimen 9.58mm (3/8 inch) thick, 88.9 x 400mm (3 1/2 x 15 3/4 inch) Homalite-100 beam with a blunt initial crack of 6.4mm (7/32 inch) in length and which was impact loaded by a drop weight of 1.48 kg (3.25 lbs) [19]. The crack emanated from the blunt saw-cut precrack and propagated through much of the height of the beam prior to curving as it approached the region of impact loading. Further details of the experimental setup, crack velocity measurements and dynamic calibration of the Homalite-100 material used are found in Reference [19]. Figure 3 shows the KJ and KJ and $\sigma_{\rm OX}$ variations obtained from the dynamic photoelastic patterns preceding and immediately after the crack curving shown in Figure 2. The negligible KJ with respect to the KJ leads to speculation that the important factor governing the crack curving is not the KJ component of the mixed mode local dynamic state of stress, but rather the $\sigma_{\rm OX}$ component which heretofore was ignored in past static analyses. The directional sta-

bility of the static Mode I crack extension with $\sigma_{\rm QX}$ as a dominant factor, however, has been presented recently [20] and thus it is conceivable that the second order term, i.e. $\sigma_{\rm QX}$ in the dynamic near-field stresses, may also govern the crack path of a rapidly propagating crack. Crack curving associated with the large positive $\sigma_{\rm QX}$ values in Fig. 3 tends to confirm this speculation.

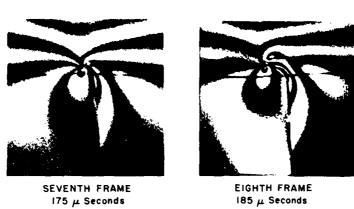


Fig. 2 Typical Dynamic Isochromatics of a Curved Crack. Homalite-100 notch bend specimen. Specimen No. 1-C042574

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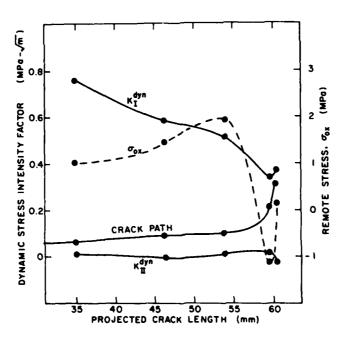


Fig. 3 Modes I and II Dynamic Stress Intensity Factors of the Curved Crack Shown in Fig. 2.

Figure 4 shows a slightly slanted crack and the associated K_{I}^{dyn} , K_{II}^{dyn} and σ_{QX} in a fracturing, wedge-loaded, double cantilever beam specimen of 9.6mm (3/8 inch) thick, 76.2 x 152.4mm (3 x 6 inch) Homalite-100 plate. Details of this experimental setup, etc., can be found in Reference [21]. Some fluctuations in K_{I}^{QY} and K_{II}^{QY} throughout the crack propagation history is noted. Again, the consistently low K_{II}^{QY} values and the large positive σ_{QX} associated with significant crack curving tend to verify the previous finding regarding the importance of σ_{QX} in dynamic crack curving. It appears then that a fracture dynamic theory comparable to that of Ref. [20] and in the presence of small but non-negligible K_{II}^{QY} may provide insight to dynamic crack curving.

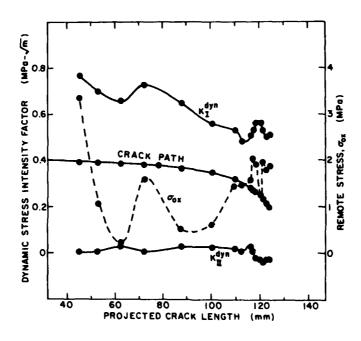


Fig. 4 Modes I and II Dynamic Stress Intensity Factors of a Slanted Crack in a Wedge-Loaded Rectangular Double Cantilever Specimen. Homalite-100 Specimen No. L31S-030274.

Figure 5 shows two dynamic photoelastic patterns of a branched crack in a single edge-notched 9.5mm (3/8 inch) thick, 254 x 254mm (10 x 10 inch) Homalite-100 plate subjected to fixed grip loading condition. Other branched cracks from this same specimen were shown in Fig. 1 and the experimental details of this test can be found in Reference [2]. As shown in Fig. 5, within the 49 micro-second interval, the propagating crack turned about 740 and arrested. The mixed Mode stress intensity factors prior to this severe crack kinking were Kinking = 0, Kinking = 0.41 MPa \sqrt{m} and $\sigma_{\rm e} = 0.18$ MPa. After crack kinking at which time the crack arrested, Kinking at Which time the crack arrested, Kinking at crack kinking can also occur under the more traditional high Kil state of stress.

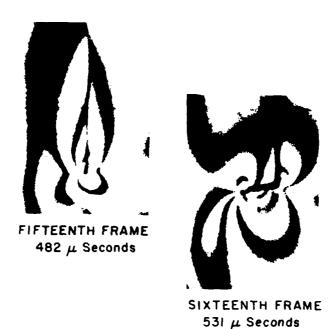


Fig. 5 Dynamic Isochromatics Prior To and After Crack Kinking. Homalite-100 Single Edge Notched Specimen Under Fixed Grip Loading. Specimen No.B5

The above three sets of data are obviously not sufficient to establish a dynamic crack curving criterion. Quantitative correlation of $K_1^{\rm dyn}$, $K_1^{\rm dyn}$ and $\sigma_{\rm dyn}$ with the degree of crack curving as well as the possible dependence on crack velocity are lacking at this time.

CRACK BRANCHING

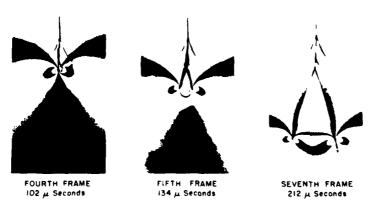


Fig. 6 Typical Crack Branching Dynamic Photoelastic Patterns Homalite-100 Single Edge Notched Specimen (Fixed Grip Loading) Specimen No. B8.

Figure 6 shows three frames out of a 16-frame dynamic photoelastic record of a crack propagating and branching in a 3.2mm (1/8 inch) thick, 254 x 254mm (10×10 inch) Homalite-100 plate loaded under fixed grip tension. Details of the experiment can be found in Reference [22].

Figure 7 shows the K^{dyn} and K^{dyn} for two branches of the cracks shown in Figure 6. By extrapolating the K^{dyn} associated with two branch cracks, an after-branching K^{dyn} = 1.2 MPa \sqrt{m} (10.90 psi $\sqrt{\text{in}}$.) and K^{dyn} = 0.45 MPa \sqrt{m} (410 psi $\sqrt{\text{in}}$) are obtained. The branching stress intensity factor, i.e. immediately prior to branching, is estimated to be K^{dyn} = 2.03 MPa \sqrt{m} (1850 psi $\sqrt{\text{in}}$).

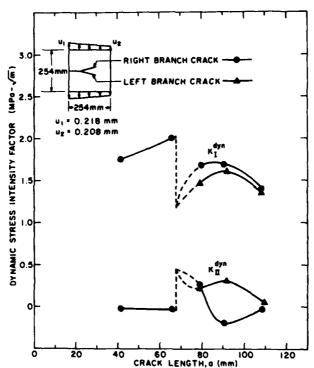


Fig. 7 Modes I and II Dynamic Stress Intensity Factors of the Branched Cracks Shown in Fig. 6.

Figure 8 shows another set of K_{I}^{dyn} and K_{I}^{dyn} for two branch cracks in a similar dynamic photoelastic experiment [20]. By extrapolating the K_{I}^{dyn} associated with the only readable right branch data, an after branching $K_{I}^{dyn} = 1.02 \text{ MPa} \ \sqrt{m} \ (920 \text{ psi} \ \sqrt{in})$ and $K_{II}^{dyn} = -0.2 \text{ MPa} \ \sqrt{m} \ (180 \text{ psi} \ \sqrt{in})$ are obtained in the extrapolated dynamic stress intensity factors prior to branching are $K_{I}^{dyn} = 2.03 \text{ MPa} \ \sqrt{m} \ (1850 \text{ psi} \ \sqrt{in})$ and $K_{II}^{dyn} = 0$.

The average branching and after-branching stress intensity factors of the above two experiments as well as those for the single experiment reported in Reference [15] yield the following: Branching K_{I}^{dyn} = 2.03 MPa \sqrt{m} (1850 psi $\sqrt{i}n$) K_{II}^{dyn} = 0 After Branching K_{I}^{dyn} = 1.0 MPa \sqrt{m} (950 psi $\sqrt{i}n$) K_{II}^{dyn} = 0.2 MPa \sqrt{m} (180 psi $\sqrt{i}n$)

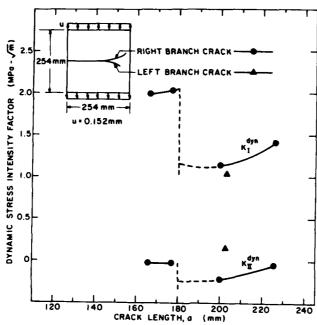


Fig. 8 Model I and II Dynamic Stress Intensity Factors of a Branched Crack in a Homalite-100 Single Edge Notched Specimen Under Fixed Grip Loading. Specimen No. B9.

The above branching K_I^{dyn} data is identical to that quoted in Reference [1]. The ratio of before over after K_I^{dyn} of 2.03/1.0 \doteqdot 2.0 is consistent with the postulate that crack branching occurs to dissipate fracture energy along two propagating cracks but is higher than the expected $\sqrt{2}$ value. It is also interesting to note that K_I^{dyn} which is prior to crack branching regains a small magnitude immediately after crack branching and is consistent with the static results of Reference [23].

DISCUSSIONS

The above dynamic photoelastic data on crack curving and crack branching should be considered as preliminary since the data is not sufficient in quantity for establishing a dynamic crack curving or a crack branching criteria. Evaluation of the accumulated dynamic photoelastic experiments using the newly developed data reduction procedure is continuing and the new crack curving and crack branching dynamic photoelastic experiments are planned.

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